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Structure-regulated Ru particles decorated P-vacancy-rich CoP as a highly active and durable catalyst for NaBH₄ hydrolysis





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HIGHLIGHTS

- Ru-cluster decorated P-vacancy-rich CoP is fabricated by a controllable method.
- The Ru_{9.8}/*r*-CoP catalyst exhibits high activity and reusability for NaBH₄ hydrolysis.
- Abundant P vacancies contribute to the formation of more uniform and smaller Ru clusters.
- The superior performance attributed to the synergy of Ru clusters and CoP support.

A R T I C L E I N F O

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G R A P H I C A L A B S T R A C T



ABSTRACT

NaBH₄ is considered the best hydrogen storage material due to its high hydrogen content of 10.6 wt% and good stability. However, NaBH₄ hydrolysis requires an efficient catalyst because of the sluggish reaction kinetics. In this work, we have demonstrated a process of preparing a cobalt phosphide-supported Ru particulate nanocatalyst with abundant phosphorus vacancies for the first time. Electron paramagnetic resonance and transmission electron microscopy revealed that the synthesized Ru_{9.8}/*r*-CoP catalyst has ample phosphorus vacancies, and Ru species are small particles (~2.5 nm) with uniform dispersion, respectively. More importantly, the optimized Ru_{9.8}/*r*-CoP catalyst has the lowest activation energy (45.3 kJ mol⁻¹) and exhibits excellent catalytic performance for NaBH₄ hydrolysis with a high hydrogen generation rate 9783.3 mLH₂ min⁻¹ g_{cat}^{-1} at 25 °C, which is higher than most of the cobalt-based catalysts. Moreover, the Ru_{9.8}/*r*-CoP catalyst also shows good reusability. For example, the catalytic performance only declined by *ca*. 14% after five cycles. The excellent catalytic performance of Ru_{9.8}/*r*-CoP is attributed to the abundant phosphorus vacancies along with a large specific surface area of *r*-CoP, which makes the Ru particles smaller and more uniformly dispersed on the surface, thereby exposing more active sites to show improved performance.

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1. Introduction

The overconsumption of fossil energy and the increasing attention to environmental issues have prompted researchers to actively seek renewable energy to replace traditional fossil fuels [1,2]. Hydrogen energy is considered to be a promising energy carrier due to its relatively high energy density, environmental friendliness, and potential use in energy conversion devices [3,4]. Among various hydrogen storage materials, NaBH₄ is considered the best hydrogen storage material due to its high hydrogen storage content of 10.6 wt%, high purity of hydrogen collected, good stability at room temperature, safe and easy to prepare [5,6]. Most importantly, recent achievements on an economical method of regenerating NaBH₄ resolved the biggest hurdle in large-scale applications [7,8]. Due to the slow selfdecomposition of NaBH₄, the hydrogen production from NaBH₄ requires efficient catalysts to promote the reaction rate. In the past, the main focus is on the noble metal-supported catalysts including Ru [9,10], Pt [11,12], Pd [13], and Rh [14] because of the high efficiency and durability. However, with the lack of natural reserves and the high cost of such catalysts, the target is gradually shifted to the synthesis of hybrid catalysts containing both noble and transition metal while keeping the similar activity and durability as pure-noble metal catalysts [15–17].

Furthermore, the metal catalysts are generally supported by silicon carbide [18,19] and carbon nanotubes [20-22]. Although the support could effectively dilute the precious metal content and lower the cost, the interaction between the two species is relatively week, resulting in poor performance, thus requires high metal concentration (~5 wt%) [23]. Recently, the first-row (3d) transition metal phosphides have attracted increasing attention because of their good activity, low cost, and multifunctionality [24–27]. Additionally, the phosphorus vacancies adjust the electronic state of metal phosphides, and regulate the interaction with noble metal particles, thereby optimizing the activation energy of the intermediate catalytic state to improve the catalyst performance [28,29]. For example, Huang et al. reported a highly efficiency CoMoP catalyst for hydrogen generation. The DFT results reveal that the outstanding HER performance results from P defect in CoMoP, preventing the active site blockage from OH* [30]. We assume that a similar approach should also positively affect the NaBH₄ hydrolysis catalyst since both catalysts involve hydrogen production and go through a similar mechanism [31]. To the best of our knowledge, we could not find a report related to this topic. As a result, we attempted to construct phosphorus vacancies in cobalt phosphide to regulate the interaction between the noble metal particles and improve its catalytic performance of NaBH₄ for hydrogen evolution, hoping to achieve a high-efficiency catalyst with a small quantity of precious metal.

In this work, we have synthesized a reduced threedimensional (3D) porous cobalt phosphide nanorods with abundant phosphorus defects by employing a facile reduction method. The effect of "killing two birds with one stone" is presented by this reduction method. The Ru loaded cobalt phosphide exhibited excellent activity and stability with a significantly low Ru loading. The crystal structure, phosphorus vacancies, microscopic morphology, and chemical state of the composites are characterized in detail. The NaBH₄ hydrolysis test shows that the designed Ru_{9.8}/*r*-CoP catalyst has a high HGR (9783.3 mL_{H2} min⁻¹ g⁻¹_{cat}), turnover frequency (450.3 mol_{H2} min⁻¹ mol⁻¹_{Ru}), and good reusability. The excellent performance is attributed to a large number of phosphorus vacancies, the high surface area provided by *r*-CoP, contributing to the high dispersion of Ru particles and the synergy between them.

2. Experimental

2.1. Preparation of rod-shaped of CoOOH

All chemicals are of analytical grade and do not require further purification. A typical procedure is as follows: $Co(CH_2COO)_2 \cdot 4H_2O$ (0.71 g), $CH_3COONa \cdot 3H_2O$ (0.82 g), and PVP (0.10 g) were added into a 100 mL glass beaker with 60 mL of deionized water. The mixture was sonicated and stirred for 30 min. Then, the resulted homogeneous solution was transferred into a 100 mL Teflonlined autoclave in an oven at 120 °C for 12 h. After that, the precipitates were collected and washed by deionized water and ethanol several times. Finally, the samples were dried in a vacuum oven at 60 °C overnight to obtain a pink powder. The possible reactions can be expressed as follows [32]:

$$\operatorname{Co}^{2+} + 2\operatorname{OH}^{-} \overset{^{\circ} C}{\to} \operatorname{Co}(\operatorname{OH})_2 \tag{1}$$

$$4Co(OH)_2 + O_2 \xrightarrow{120 \circ C} 4CoOOH + 2H_2O$$
(2)

2.2. Preparation of CoP

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In a typical preparation of CoP, the CoOOH powder (0.08 g) was located in the middle of a quartz tube with NaH₂PO₂ (1.0 g) at the upstream side near the CoOOH precursor. The sample was subsequently heated to 350 °C with a rate of 5 °C min⁻¹ and kept for 2 h under a N₂ atmosphere. After the phosphidation, the samples were cooled down to ambient temperature in flowing N₂ gas. The possible reactions are described as follows [33]:

$$4NaH_2PO_2 \xrightarrow{350 \ C} Na_44P_2O_7 + 2PH_3 + H_22$$
 (3)

$$CoOOH + PH_3 \xrightarrow{350 \circ C} CoP + 2H_2O$$
(4)

2.3. Preparation of reduced cobalt phosphide

Briefly, 0.10 g CoP was dispersed in 40 mL 1.5 M NaBH₄ solution (or 1.0 M, 2.0 M, and 0.08 M NaBH₄ solutions) and stirred at room temperature for 2 h. Then, the products were collected and washed by deionized water and ethanol several times, finally, the samples were dried in a vacuum oven at 60 °C overnight and marked as *r*-CoP (or r_1 -CoP × r_2 -CoP and r_3 -CoP).

2.4. Preparation of Ru-species/r-CoP

Typically, the *r*-CoP (30.0 mg) was added into a 50 mL glass beaker with 20.0 mL of deionized water and 12.0 mg (or 6.0, 9.0, 15.0 and 18.0 mg) of RuCl₃. The mixed solution was stirred at room temperature for 4 h, then 5 mL (0.08 M) NaBH₄ solution was slowly added dropwise, and stirring continued for 0.5 h. The products were collected and washed by deionized water and ethanol several times. Subsequently, the samples dried in a vacuum oven at 60 °C for 12 h. The catalysts are labeled according to the Ru species' actual contents determined by ICP-AES (Table S1). For instance, the Ru_{9.8}/*r*-CoP represents the sample with 9.8 wt% Ru. The Ru content was altered by changing the RuCl₃ amounts in the precursor.

2.5. Catalytic hydrolysis of NaBH₄ measurements

The catalytic performance of catalytic materials is evaluated by the following test methods: To a 50 mL mixed solution containing 150 mM NaBH₄ and 0.4 wt% NaOH was added 10 mg catalyst under vagrous stirring at 25 °C. The classical water-displacement technique is used to measured the generated H₂, and the drainage system is connected to a computer to record instantaneous water volume changes. During the reusability study, the catalyst was collected by centrifugation and dried. The collected catalyst from the previous experiment was added to the fresh NaBH₄ solution in the next experiment. The catalyst's activation energy was tested in the temperature range from 298 to 318 K, and then calculated by the Arrhenius equation. All tests are repeated three times to ensure reliable results.

The hydrogen generation rate (HGR) and turnover frequency (TOF) values are calculated according to the following equations [34,35],

$$HGR = \frac{V_{H_20}(mL)}{t(\min) \times m(g)}$$
(5)

$$TOF = \frac{n_{H_2}(mol)}{t(\min) \times n_{Ru}(mol)}$$
(6)

where V_{H2O} is the volume of drained water, *m* is the total mass of the catalyst, n_{H2} is the moles of generated H₂, n_{Ru} is the moles of Ru in the catalyst, and *t* is the total reaction time in minutes.

$$\ln \kappa = \ln A - E_a / RT \tag{7}$$

Equation (7) is the Arrhenius equation, where κ (L min⁻¹ g⁻¹) is the hydrogen evolution slope, A (L min⁻¹ g⁻¹) is a constant, E_a (kJ mol⁻¹) is the activation energy, R (J mol⁻¹ K⁻¹) is the gas constant, and T (K) is the temperature of the during the test.

3. Results and discussion

The synthetic processes of the Ru/*r*-CoP are illustrated in (Fig. 1a). The CoOOH nanorods were first synthesized by low-temperature hydrothermal methods using cobalt acetate and sodium acetate as precursors combined with PVP as a soft template. The CoOOH precursor template was then placed in a quartz boat and phosphorized in a nitrogen atmosphere at a relatively low temperature of 350 °C. The P vacancies were generated by dispersing the as prepared CoP nanorods into NaBH₄ solution (1.5 M). The highly reductive condition generates *r*-CoP with a large number of phosphorus vacancies. The *r*-CoP was redispersed in RuCl₃ solution, and then the NaBH₄ solution (0.08 M) was added dropwise to reduce the Ru³⁺ to form Ru/*r*-CoP.

As shown in (Fig. 1b), the nanorod-shaped CoOOH was synthesized by low-temperature hydrothermal method (the inset is the SEM with different magnifications). After phosphorized, the morphology of CoP still maintained the original rod-like characteristics (Fig. 1c). However, the surface of *r*-CoP became rough and porous (Fig. 1d) after treatment with NaBH₄ solution. As shown in Fig. 1e and f, the Ru/r-CoP and r-CoP show similar morphology and the Ru particles are uniformly dispersed on the surface of r-CoP (the red circle indicates the position of the Ru particles). The statistical results of about 200 Ru particles concluded that the average particle size is approximately 2.52 nm (Fig. S1 and inset Fig. 1f). In contrast, the Ru particles show an apparent agglomeration on Ru_{9.8}/CoP (Fig. 1g). The result indicates that higher surface area and porosity help to disperse the Ru ions thereby prevent the agglomeration during reduction. The high-resolution TEM (HRTEM) image (Fig. 1h) further confirmed the crystal structure of the $Ru_{9,8}/r$ -CoP whereas the CoP nanorods show two clear lattice spacing of 0.20 and 0.37 nm, corresponding to the crystal planes of (210) and (101) [36]. Besides, the other two lattice fringes at 0.22 and 0.19 nm are attributed to the (210) and (101) crystal planes of the chemically deposited RuO₂ species, respectively [37] indicating the Ru particles were oxidized upon exposure to the air. The highangle annular dark-field (HAADF) TEM mapping shows that Ru, Co,

P, and O elements are evenly distributed throughout the $Ru_{9.8}/r$ -CoP (Fig. 1i).

X-ray diffraction (XRD) pattern is used to explore the crystal structure of different composite materials. As shown in Fig. S2, the crystal structure of the Co-species prepared by the hydrothermal method is consistent with the standard CoOOH diffraction peak [38]. After phosphorization (Fig. 2a), the XRD patterns of the prepared sample are well-matched with the typical CoP diffraction peaks (JCPDS: 29-0497) [39]. Meanwhile, the *r*-CoP diffraction peaks are consistent with CoP and indicate that a NaBH₄ reduction does not affect the crystal structure of the material. Notably, the XRD peak intensity of Ru_{9.8}/*r*-CoP becomes weaker after the reduction and Ru loading. Fig. 2b shows five-strong Raman scattering peaks at 190, 467, 510, 601, and 668 cm⁻¹ of all the samples, corresponding to the distinct regions of cobalt oxide, which may be caused by the oxidation of CoP in the air [40].

Furthermore, it is reported that P vacancies generated by NaBH₄ treatment [30,31] accelerate the electron transport during the reaction, thereby accelerating the hydrogen evolution process of NaBH₄ [41]. We first studied the effect of different concentrations of NaBH₄ on the generation of P vacancies. EPR spectrum found that the most substantial unpaired electron peak was produced when NaBH₄ concentration was 1.5 M, suggesting that the largest proportion of P vacancies can be generated at g = ~2.08 (Fig. S3). After the Ru-species were modified (Fig. 2c), the EPR intensity was somewhat weakened, which may be due to the Ru-species occupying some of the P defect sites.

The BET specific surface area of $Ru_{9.8}/r$ -CoP, r-CoP, and CoP are calculated to be 71.1, 39.6, and 6.6 m² g⁻¹ (Fig. 2d), along with the adsorption average pore diameters are about 3.48, 3.51, and 2.24 nm, respectively [42]. Compared with CoP, the r-CoP has a larger BET surface area and porosity, which can disperse Ru particles more uniformly. The $Ru_{9.8}/r$ -CoP catalyst exhibits the largest BET surface area resulting in smaller Ru particles and more uniform dispersion, consequently providing more active sites and accelerating the transport of solutes and gas emissions compared to other catalysts, thereby improving catalytic performance [43,44].

To understand the catalyst's electronic state changes, we performed XPS characterization (Fig. 3). The XPS full spectra of the catalysts in Fig. S4 indicate the presence of C, O, P, Co, and/or Ru elements, which are consistent with theoretical values. The highresolution XPS spectra of C 1 s + Ru 3d regions (Fig. 3a) are convoluted into C-C (284.8 eV) and C-O (286.0 eV) used as calibration standards [45,46], and the Ru 3d core levels from Ru_{9.8}/r-CoP and Ru_{9.8}/CoP are convoluted at the binding energies of 281.1 eV, 282.3 eV (Ru 3d_{5/2}) and 285.5 eV, 286.7 eV (Ru 3d_{3/2}) [47,48]. It can be seen that the area ratio of RuO₂ in r-CoP is increased to 73.7% by comparison with CoP (28.2%). The catalyst with a higher RuO₂ content showed a better reaction rate similar to that reported in the recent literature [49,50], revealing that the NaBH₄ hydrolysis reaction-rate could be improved significantly through the synergy between the Ru and RuO₂ species in the catalyst. As shown in Fig. 3b, the Co 2p_{3/2} regions' peaks at 778.7, 781.6, and 787.1 eV can be ascribed to the Co-P, Co-O, and satellite peak, respectively [51–53]. Notably, except for CoP that has an obvious Co-P peak, all other catalysts' Co-P peaks become so weak that they are not detectable. This result may be because the oxide layer (r-CoP, Ru_{9.8}/CoP, and Ru_{9.8}/r-CoP) formed on the catalyst's surface that is thicker than the penetration depth of XPS (about 10 nm). It should be emphasized that we can see clear CoP characteristic peaks in the XRD diffraction spectrum (Fig. 2a), which further proves that only the sample surface is oxidized. The existence of such an oxide layer may be one of the reasons for the formation of more homogeneously dispersed RuO₂ species on the surface of r-CoP. The P 2p regions' peaks at 130.29 and 133.79 eV can be



Fig. 1. (a) Fabricating procedure of Ru/*r*-CoP. SEM images of (b) CoOOH, (c) CoP, (d) *r*-CoP, and (e) Ru_{9.8}/*r*-CoP. TEM images of (f) Ru_{9.8}/*r*-CoP and (g) Ru_{9.8}/*CoP*. HRTEM images of (h) Ru_{9.8}/*r*-CoP and elemental mappings of (i) Ru, Co, P and O.

ascribed to the Metal–P and P–O peak (Fig. 3c), respectively [54,55].

The H₂ production of NaBH₄ hydrolysis using the prepared catalysts was carried out in an alkaline NaBH₄ solution at 25 °C. The schematic diagram of the catalytic hydrolysis device is shown in Fig. S5. Where the amount of H₂ produced is calculated based on the weight of the water replaced by H₂. The self-hydrolysis of NaBH₄ has also been studied. When only 150 mM NaBH₄ solution is present, very little H₂ gas is produced (Fig. S6). However, when an additional 0.4 wt% NaOH is introduced, almost no gas is detected, indicating that NaBH₄ is very stable under an alkaline environment (Fig. S7). Moreover, the Ru content of different catalysts was determined by ICP-AES (Table S1), which were 4.5, 5.7, 9.8, 12.5, and 13.4 wt%, respectively. The correlation between the amounts of hydrogen produced with Ru loadings is shown in Fig. 4a. The change of HGR is volcanic. When the content of Ru is 9.8 wt%, the HGR has a maximum value of 9783.3 mL min⁻¹ g_{cata-} l_{vst}^{-1} , and the TOF is 450.1 mol min⁻¹ mol_{Ru}⁻¹ (Fig. 4b), which is one of the highest TOF and HGR currently reported in the literature (Table S3). The excellent catalytic performance can be attributed to the abundant phosphorus vacancies, uniformly dispersed the Ru-species, thereby exposing more active sites. To explore the influence of NaBH₄ concentration on the rate of hydrogen evolution, we used different gradient concentrations of NaBH₄. Fig. 4c shows that with the increase of NaBH₄ concentration, the HGR is almost unchanged. The NaOH concentration has a minimum effect on the HGR after a certain level [56]. We observed similar phenomena in this work (Fig. 4d), signifying the stabilizing rules of NaOH [7,57].

To further study the influence of various catalysts on the rate of hydrogen evolution, CoOOH, CoP, r-CoP, Ru_{9.8}/CoP, and Ru_{9.8}/r-CoP were investigated. As shown in Fig. 5a and b, CoOOH, CoP, and r-CoP show low hydrolysis activity towards NaBH₄ (Note: These TOF values are calculated from the Co content obtained by the ICP-AES test as shown in Table S2). When the Ru loading is 9.8 wt%, Ru_{9.8}/r-CoP shows the highest hydrogen evolution rate and the highest TOF. The results show that loading a small amount of Ru can significantly increase the HGR of NaBH₄ hydrolysis. To measure the activation energy, a set of experiments were conducted at different temperatures using the $Ru_{9.8}/r$ -CoP along with CoP. As shown in Fig. 5c and Fig. S8, as the reaction temperature increases from 298 K to 318 K, the generation rate of H₂ increases rapidly. The rate constant (k) is calculated from the initial slope of each experiment. As shown in Fig. 5d, the activation energy of $Ru_{9.8}/r$ -CoP catalyst was calculated to be 45.3 kJ mol⁻¹ by the Arrhenius equation ($\ln k vs. 1/T$), much lower than CoP (56.5 kJ mol⁻¹), which implies that Ru has an enhanced reaction activity to the hydrolysis of NaBH₄ to produce H₂ [58].

To evaluate the catalyst's reusability, we assessed the optimized $Ru_{9.8}/r$ -CoP catalyst's stability through a continuous cycle experiment of NaBH₄ hydrolysis in an alkaline solution. After each cycle, the catalyst was washed, freeze-dried, and weighed in sequence. The results show that after five consecutive cycles, the catalytic performance of the $Ru_{9.8}/r$ -CoP is slightly decayed (Fig. 6a and b).



Fig. 2. (a) XRD pattern of CoP, *r*-CoP and Ru_{9.8}/*r*-CoP. (b) Raman spectra of CoP, *r*-CoP, Ru_{9.8}/CoP and Ru_{9.8}/*r*-CoP. (c) EPR spectra of CoP, *r*-CoP, Ru_{9.8}/CoP and Ru_{9.8}/*r*-CoP. (d) Nitrogen adsorption-desorption isotherms of CoP, *r*-CoP and Ru_{9.8}/*r*-CoP.



Fig. 3. (a) C 1s and Ru 3d XPS spectra of Ru_{9.8}/CoP and Ru_{9.8}/r-CoP. (b) Co 2p XPS spectra of CoP, r-CoP, Ru_{9.8}/CoP and Ru_{9.8}/r-CoP. (c) P 2p XPS spectra of CoP, r-CoP, Ru_{9.8}/CoP and Ru_{9.8}/r-CoP.

After five cycles, the catalyst remained intact (Fig. S9) and kept the same crystal structure (Fig. S10). However, there is a slight reduction in TOF. However, part of the catalyst deactivation can be attributed to Ru-species exfoliation (Fig. S11) and catalyst poisoning caused by BO_2^- species [59].

The catalytic mechanism diagram of the hydrolysis of NaBH₄ solution to hydrogen is proposed in Fig. 6c. The presence of phosphorus vacancies promotes the smaller and more uniformly dispersed Ru particles, consequently increase the grain boundary between the Ru particles and the transition metal phosphide support. Due

to the charge transfer from CoP species to Ru resulted from the work function differences (Note: the work functions of Ru is 4.71 eV and CoP is 4.227 eV) [60,61], the Ru becomes electron-rich, thereby selectively binds with B of the BH_4^- ion (Fig. 6c) [62]. Meanwhile, the water is adsorbed by the CoP surface through the P vacancies' binding sites. The adsorbed water molecules can interact with the Ru-BH₄⁻ at the interface to release one molecule of H₂ and form a BH₃OH intermediate [63,64]. As the reaction proceeds, the rest of the H atoms in the borohydride are replaced by OH⁻ ions, ultimately dissociating the B(OH)₄⁻ species [62].



Fig. 4. (a) The relationship between H_2 evolution and loadings of Ru species in Ru_x/r -CoP (*x* is the mass percentage of Ru) catalysts, and (b) the summarized TOF values and HGR. The influence of $Ru_{9.8}/r$ -CoP at (c) different NaBH₄ concentrations and (d) different NaOH percentages on NaBH₄ hydrolysis to produce H_2 .



Fig. 5. (a) Different catalyst stoichiometric H₂ evolution, and (b) the corresponding TOF and HGR of the catalyst summarized from a. (c) Stoichiometric H₂ evolution of Ru_{9.8}/r-CoP at different reaction temperatures, and (d) the calculated activation energies of Ru_{9.8}/r-CoP and CoP catalysts.



Fig. 6. (a) The stoichiometric H₂ release of the cycle stability test of Ru_{9.8}/*r*-CoP, (b) the corresponding TOF values from a. (c) The proposed catalytic mechanism diagram of Ru_{9.8}/*r*-CoP catalyst generating H₂ by hydrolysis of NaBH₄ solution.

4. Conclusions

In summary, we have successfully developed a facile strategy to prepare Ru particles loaded on *r*-CoP nanorods that produce H₂ by the hydrolysis of NaBH₄ in an alkaline solution. EPR, XPS, and TEM analysis of Ru_{9.8}/r-CoP material showed that Ru particles were uniformly dispersed on r-CoP nanorods' surface, thereby creating channels for rapid electron transfer, and full exposure of active sites. The superior performance of Ru_{9.8}/r-CoP mainly stems from the following facts. Firstly, the reduced cobalt phosphide has a porous 3D nanorod structure, enhanced BET surface area, and specific pore structure, which provides a good platform for the uniform dispersion of Ru particles. Secondly, the *in-situ* generated Ru particles on the *r*-CoP surface have a smaller particle size and expose more active sites induced by P vacancies, thereby obtaining higher catalytic activity. This research provides a new way to develop metal phosphide based a high-performance and low-loading noble metal catalyst for NaBH₄ hydrolysis.

CRediT authorship contribution statement

Shuqing Zhou: Writing - original draft. Yuting Yang: Investigation. Wanyu Zhang: Data curation. Xianfa Rao: Data curation. Puxuan Yan: Investigation. Tayirjan Taylor Isimjan: Supervision. Xiulin Yang: Supervision, Writing - review & editing.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.jcis.2021.02.009.

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