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Fine-tune *d*-band center of cobalt vanadium oxide nanosheets by *N*-doping as a robust overall water splitting electrocatalyst



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HIGHLIGHTS

- A serial of *N*-doping cobalt vanadium oxide electrocatalysts were designed by two steps.
- Nitride temperature affects the microstructure, crystallinity and N doping content of the catalyst.
- *N*-doping cobalt vanadium oxides display excellent activity and stability for OER and HER.
- DFT calculations suggest that *N*doping can fine-tune the *d*-band center and band gap to facilitate intermediate adsorption and electron motion.

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G R A P H I C A L A B S T R A C T

A series of cobalt vanadium oxides were synthesized by feasible hydrothermal growth followed by nitridation at different temperatures. N-Co₂V₂O₇@NF (350 °C) and N-Co₂VO₄/VO₂@NF (400 °C) exhibited excellent OER and HER performance with overpotentials of 310 mV and 224 mV at 100 mA cm⁻², long-term stability for 170 h and 700 h at 100 mA cm⁻², and assembled as anode and cathode for robust overall water splitting (1.93 V, 500 mA cm⁻²).



ABSTRACT

Developing high-activity, long-durability, and noble metal-free oxygen evolution (OER) and hydrogen evolution (HER) cocatalysts are the bottlenecks for efficient overall water splitting (OWS). Here, novel cobalt vanadium oxides doped by nitrogen were synthesized by nitriding $C_0_2V_2O_7@NF$ precursor at 300–450 °C for OER and HER reactions. N- $C_02V_2O_7@NF$ (350 °C) and N- $C_02VO_4/VO_2@NF$ (400 °C) show remarkable OER and HER performance with overpotentials of 310 mV and 224 mV at high current density (100 mA cm⁻²). Besides, they also revealed long-term solid stability even after 170 h and 700 h of continuous performance. Furthermore, the N- $C_02V_2O_7@NF^{(+)}||N-C_02VO_4/VO_2@NF^{(-)}$ OWS device possesses a cell voltage of 1.93 V at 500 mA cm⁻² better than $RuO_2@NF^{(+)}||Pt/C@NF^{(-)}$ (2.02 V) and can operate for 60 h with almost no degradation. This extraordinary performance can be attributed to the nanosheet structure, which can maximize the exposure of active sites and accelerate the mass transfer. Moreover, density functional theory (DFT) calculations suggest that *N*-doping can fine-tune the d-band center





and band gap to facilitate intermediate adsorption and electron motion. The method presented here can be applied in other novel *N*-doped electrocatalysts for the energy field.

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1. Introduction

As an alternative to conventional fossil fuels, electrochemical overall water splitting (OWS) is a viable method for producing green hydrogen [1]. However, the high overpotentials of oxygen and hydrogen evolution cocatalysts (OER and HER) limit the system efficiency [2]. RuO₂ and Pt/C are the most potent OER and HER catalysts reported so far, but their rarity and high-cost limit further utilization at the industrial scale [3]. Thus, it is of industrial significance to explore the OER and HER catalysts with low-cost, high-activity and long-term durability for water splitting.

Recently, abundant reserves of transition metal-based materials have been reported for OER and HER, including oxides [4], phosphides [5], and nitrides [6,7]. Among them, transition metal oxides have been widely investigated owing to good stability in alkaline media [8]. Especially, Cobalt vanadium oxide showed an interesting property where the V atom shares *d* electrons to promote the metallic behavior [9]. However, this process is restricted by oxides' low instinctive electronic conductivity [10,11]. As a result, increasing the conductivity of metal oxide is necessary for better performance, which can be improved by *N*-doping [12,13]. Because the N-doping causes uneven spin density and charge distribution around adjacent atoms, enhancing absorption of reaction intermediates, thereby optimizing OER and HER performance [14,15]. Up to date, fair amounts of studies focusing on N-doped Mo- [16], Co- [17], Ni-based [18], sulfides [19-21], oxides [22] and carbides [23.24] were reported. Nevertheless, the HER or OER performance of the reported materials is still inadequate. On the one hand, their performances are far from Pt/C and RuO₂ catalysts [25]. On the other hand, they do not have sufficient stability in the alkaline environment at high current densities (>100 mA cm⁻²), hindering their industrial application [26]. These limitations could be overcome by the *N*-doping in cobalt vanadium oxide to enhance the conductivity, thereby improving the HER and OER activities.

Here, we reported a series of N-doped cobalt vanadium oxides by a facile nitrification process in NH₃. The material exhibited a crystal form of $Co_2V_2O_7$ below 350 °C and would transform into Co_2VO_4/VO_2 at 400 °C or Co_2VO_4/V_2O_3 at 450 °C, respectively. Meanwhile, as the nitrification temperature increased, the nanosheet morphology of the samples gradually became rougher and thicker, affecting the exposure of active sites. Once pyrolysis temperature exceeds 350 °C, the N doping content reaches a maximum value (~2 %). The resulting N-Co₂V₂O₇@NF (350 °C) and N-Co₂VO₄/VO₂@NF (400 °C) can be used for OER or HER to achieve overpotentials of 310 mV and 224 mV at 100 mA cm⁻² to stable for 170 h and 700 h, respectively. Density function theory (DFT) indicates that N-doping can distinctly regulate the material's band gap, d-band center and absorption/desorption energy of intermediates. Finally, the N-Co₂V₂O₇@NF⁽⁺⁾||N-Co₂VO₄/VO₂@NF⁽⁻⁾ OWS system exhibited a cell voltage of 1.93 V at 500 mA cm⁻² being superior to RuO₂@NF⁽⁺⁾||Pt/C@NF⁽⁻⁾ (2.02 V) and can maintain for 60 h with only 1.8 % decline.

2. Experimental section

2.1. Materials

Cobalt chloride hexahydrate (CoCl₂·6H₂O, 99.5 %, Aladdin, 500 g), ammonium metavanadate (NH₄VO₃, 99.0 %, Aladdin,

100 g), potassium hydroxide (KOH, 90 %, Macklin, 500 g), ethanol (C₂H₅OH, 99.7 %, Xilong, 500 mL), hydrochloric acid (HCl, 37 %, Xilong, 500 mL), commercial Pt/C (20 wt% Pt, Sinero, 1 g), RuCl₃-·3H₂O (37 %, Inno-chem, 25 g), Nafion solution (5 wt%) and commercial Pt/C (20 wt% Pt, Sinero, 1 g) were purchased from Alfa Aesar. Nickel foam (NF, thickness: 1.6 mm) was obtained from a commercial source. Deionized water was used in the experiments. All chemical reagents were used directly without further purification.

2.2. Synthesis of Co₂V₂O₇@NF catalysts

A piece of NF ($1.5 \times 3.5 \text{ cm}^2$) was washed with 1 M HCl, distilled water and ethanol under ultrasonication for 15 min, respectively, to eliminate surface impurities. Typically, 0.8 mmol CoCl₂·6H₂O and 0.8 mmol NH₄VO₃ were dissolved into 30 mL distilled water with persistently stirring at 70 °C for 10 min, respectively. After cooling naturally, above two pieces of solutions were mixed with ultrasonication for 1 h and transferred into a 100 mL autoclave with pretreated NF to heat at 180 °C for 18 h. After cooling, the black NF was taken out, rinsed with distilled water several times and dried overnight (mass loading: 2.0 mg cm⁻²).

2.3. Synthesis of a serial of N-doping cobalt vanadium oxide catalysts

The as-produced Co₂V₂O₇@NF was first loaded into a porcelain boat and inserted into reaction quartz tube to fix in tube furnace. Afterward, the N₂ was introduced for 15 min to exclude the air in this system. Subsequently, NH₃ was flowed into at 300 °C, 350 °C, 400 °C, and 450 °C for 3 h (2 °C min⁻¹), respectively. Waiting until the temperature drops to circumstance temperature, the deep-black produced NF was taken out and named as N-Co₂V₂O₇@NF-300, N-Co₂V₂O₇@NF, N-Co₂VO₄/VO₂@NF and Co₂VO₄/V₂O₃@NF, respectively (mass loading: 2.1, 2.0, 2.1, and 1.8 mg cm⁻², respectively). As a comparison, Co₂V₂O₇@NF-350 was synthesized through pyrolysis of Co₂V₂O₇@NF at 350 °C for 3 h under an Ar atmosphere.

2.4. Synthesis of RuO₂ and 20 wt% Pt/C electrodes

The RuCl₃·3H₂O was directly calcined at 400 °C for 3 h in the air to obtain RuO₂ powders. Then, 2 mg RuO₂ or 20 wt% commercial Pt/C was dispersed into a mixture of 100 μ L deionized water, 100 μ L ethanol and 5 μ L 5 wt% Nafion, respectively. The mixture was ultrasonically treated for at least 30 min to form a uniform catalyst ink, then dropped onto the surface of NF (1 \times 1 cm²) and dried naturally in the air. The corresponding XRD pattern is indexed to RuO₂ (JCPDS: 40–1290), confirming the successful preparation of RuO₂. The SEM image indicates that RuO₂ exists in the form of nanoparticles (Fig. S2).

2.5. Electrochemical tests

Electrochemical tests were conducted in Biologic VMP3 electrochemical workstation with a typical three-electrode system in 1.0 M KOH electrolyte. The catalyst ($1 \times 1 \text{ cm}^2$), graphite plate and saturated calomel electrode (SCE) were used as working, counter and reference electrode, respectively. Cyclic voltammetry (CV) measurement was performed with a sweep rate of 5 mV s⁻¹ at potentials of -0.9 to -1.5 V and 0 to 0.8 V for HER and OER, respectively. Linear sweep voltammetry (LSV) was recorded with a scan rate of 0.2 mV s⁻¹ and electrochemical impedance spectroscopy (EIS) test was conducted near the onset potential in the frequency range from 100 kHz to 10 mHz. The electrochemical double layer capacitance (C_{dl}) was obtained by CV tests in a nonfaradaic region with scan rates from 2 to 12 mV s⁻¹. Chronopotentiometry measurements achieved the long-term stability test of the catalysts at a current density of 100 mA cm⁻². Additionally, the CV was performed with a sweep rate of 5 mV s⁻¹ at a potential of 0 to -1.5 V (HER) and 0 to 0.8 V (OER), respectively. All potentials of the reversible hydrogen electrode (RHE) were calibrated by the equation: $E_{RHE} = E_{SCE} + 0.241 + 0.059 \times pH$ (Fig. S1). The OWS test was performed in 1.0 M KOH electrolyte in a potential range of $0 \sim 2.5$ V (vs SCE) with a scan rate of 5 mV s⁻¹.

2.6. Characterization

A scanning electron microscope (SEM, FEI quantum 200 FEG) and transmission electron microscope (TEM, JEM-2100F) characterized the sample's microstructure. X-ray powder diffraction (XRD, D/Max-3c at 12 kW with Cu K α radiation) patterns were obtained using a scanning speed of 5.0° min⁻¹. The chemical valence states of different elements were certified by X-ray photoelectron spectroscopy (XPS, JPS-9010 Mg Ka). Inductively coupled plasma atomic emission spectrometry (ICP-AES, IRIS Intrepid II XSP) was used to verify the actual loading of different metals in catalysts. The Raman spectrum comes from an in Via confocal Raman microscope (Renishaw, UK). The specific surface area and pore size distribution were calculated according to the Brunauer-Emmett-Teller (BET) and t-Plot methods, respectively. The nitrogen content is obtained through an element analyzer (Vario EL III). We performed multiple experiments using elemental analyzers, XPS, and ICP-AES to reduce errors and ensure data accuracy and reproducibility.

3. Result and discussion

N-doped cobalt vanadium oxides sequence was synthesized through a feasible two-step process (Fig. 1a). Initially, Co₂V₂O₇ nanosheet precursor was prepared via hydrothermal deposition of CoCl₂ and NH₄VO₃ at 180 °C, and the X-ray diffractograms (XRD) is assigned to Co₂V₂O₇ (JCPDS: 38–0252). After that, N elements were introduced into Co₂V₂O₇ nanosheets along with a phase transformation through pyrolysis at 300 \sim 450 °C. When the temperature is lower than 350 °C, the Co₂V₂O₇ phase persists and converts into Co_2VO_4/VO_2 and Co_2VO_4/V_2O_3 at 400 $\,^\circ C$ and 450 °C, respectively, which can index well to those of standard cards: Co₂V₂O₇ (JCPDS: 38-0252), Co₂VO₄ (JCPDS: 73-1633), VO₂ (JCPDS: 81-2392) and V₂O₃ (JCPDS: 71-0280) (Fig. 1b and S3). Raman spectrum of N-Co₂V₂O₇ demonstrates two prominent peaks at 507 and 617 cm⁻¹, corresponding to Co-N and V-N bonds (Fig. 1c) [12,27]. The remaining two located at 266 cm⁻¹ and 400 cm^{-1} is identified as V–O and Co–O bonds, respectively [28,29]. Notably, V–O bands of $Co_2V_2O_7$ shift from 277 cm⁻¹ to 266 cm⁻¹ (N-Co₂V₂O₇), indicating that the V-O bond strength change after N doping [30,31]. Brunauer–Emmett–Teller (BET) N₂ adsorption-desorption isotherm displays a typical V-type hysteresis loop, and N-Co₂V₂O₇ possesses a mesoporous structure with pore sizes of around 17.9 nm (Fig. S4). In a word, the N element was successfully doped into the Co₂V₂O₇ precursor, and the crystal phase of *N*-doped cobalt vanadium oxide strongly depends on the nitrification temperature.

SEM and TEM were used to examine the morphology. After the hydrothermal reaction, $Co_2V_2O_7$ precursor nanosheets densely

grow on the NF surface (Fig. 1d). The spacing between nanosheets increased gradually without change of morphology (thick: $5 \sim 7$ nm) at low nitrification temperature (350 °C), which facilitates active site exposure and gas release (Fig. 1e and S5-7a) [32]. Dramatically, the corresponding nanosheet structures became thicker and eventually aggregated together at higher temperatures (400 \sim 450 °C), indicative of the products' thermal expansion and chemical composition changes (Fig. S7b-c). Note that the Co₂V₂O₇@NF-350 material as a control showed almost no noticeable change in morphology after calcination in Ar at 350 °C (Fig. S8). TEM image reveals the well-defined nanosheet structure of $N-Co_2V_2O_7$ and $N-Co_2VO_4/VO_2$, according to the SEM results (Fig. 1f). In the HRTEM image, interplanar spacings of 0.324 nm, 0.312 nm, and 0.175 nm meet with the (112), (002) and (204) crystal planes of Co₂V₂O₇, respectively (Fig. 1g and Fig. S9a). Meanwhile, the HRTEM image of $N-Co_2VO_4/VO_2$ shows that interplanar spacings of 0.615 nm and 0.296 nm correspond to (001) and (220) crystal planes of VO₂ and Co₂VO₄ (Fig. S9b), respectively. Moreover, the observable interface between the VO₂ and Co₂VO₄ crystal phases indicates the existence of heterostructures. SAED image of N-Co₂V₂O₇ reconfirms that the sample possesses a single crystal structure, and the (-111), (-212) and (-112) crystal planes correspond to $Co_2V_2O_7$, respectively (Fig. 1h). HAADF-STEM and corresponding element mapping images reveal that Co, V, O, and N are evenly distributed on the nanosheet (Fig. 1i). Findings mentioned above certify that the N element was triumphantly doped into the sheet-shaped Co₂V₂O₇ precursor and can modify the product's morphology under various temperatures.

The chemical states of cobalt vanadium oxides were further explored through XPS. All substances contain obvious Co 2*p*, V 2*p* and O 1 *s* peaks in the full spectrum (Fig. S10). The N content in N-Co₂V₂O₇-300 is only 1.0 % from the elemental analyzer, yet in N-Co₂V₂O₇-350, N-Co₂VO₄/VO₂-400 and N-Co₂VO₄/V₂O₃-450 are 2.0 %, 2.1 % and 2.1 % without rising apparently, respectively (Table S1). Notably, a small amount of N (about 0.1 %) is present in Co₂V₂O₇ due to the NH₄VO₃ precursor, which is insufficient to change the catalytic performance (Table S1 and Fig. S11).

In the Co 2p XPS spectra, four peaks are observed at 782.3 eV $(Co^{2+} 2p_{3/2})$, 798.1 eV $(Co^{2+} 2p_{1/2})$, 780.6 eV $(Co^{3+} 2p_{3/2})$ and 796.5 eV ($Co^{3+} 2p_{1/2}$) (Fig. 2a), and the content of Co^{2+} is increased slightly with the rising nitrification temperature, which can be ascribed to the reduction properties of NH₃ (Table S2) [33]. V 2p spectra display-two doublet peaks at 516.2 eV/523.6 eV and 517.1 eV/524.5 eV, belonging to V^{4+} and V^{5+} , respectively (Fig. 2b) [34,35]. As the nitridation temperature reaches 400 °C, V^{5+} disappears, while V^{3+} appears at the binding energy (515.2 e V/522.6 eV), implying N-Co₂V₂O₇ is converted into N-Co₂VO₄ and VO₂ [36]. In the N 1 s spectrum, apart from the N-metal (N-M) bond (397.1 eV) formed by N-doping into the oxide, the peak at 399.6 eV originates from the N - H surface terminal group after NH₃ treatment (Fig. 2c) [37-39]. In short, the high-temperature environment can promote N doping into oxide precursors and induce cobalt vanadium oxide's reconstitution.

The catalytic activities of different materials were measured in a 1.0 M KOH solution. As shown in Fig. 3a, three *N*-doped cobalt vanadium catalysts with about 2 % *N*-doping content exhibit catalytic performance comparable to RuO₂. Specifically, N-Co₂V₂O₇@NF can achieve the smallest overpotentials of 244 mV, 287 mV and 310 mV to afford current densities of 10, 50, and 100 mA cm⁻², respectively, which is visibly better than other catalysts (Fig. S12). Tafel slopes were calculated and displayed in Fig. 3b to understand the reaction mechanism better. The N-Co₂V₂O₇@NF presents the smallest Tafel slope of only 56.0 mV dec⁻¹ among all catalysts, indicating the fastest OER reaction kinetics [40]. Compared with most recently reported OER electrocata-



Fig. 1. (a) Schematic illustration of preparing *N*-doped cobalt vanadium oxides on NF. (b) X-ray diffractograms (XRD) and (c) Raman spectra of Co₂V₂O₇ and N-Co₂V₂O₇. (d-e) Scanning electron microscopy (SEM) images of Co₂V₂O₇@NF and N-Co₂V₂O₇@NF. (f) Transmission electron microscopy (TEM), (g) high-resolution TEM (HRTEM), (h) selected area electron diffraction (SAED) pattern, and (i) high angle annular dark-field scanning transmission electron microscope (HAADF-TEM) image and corresponding elemental distribution of N-Co₂V₂O₇@NF.



Fig. 2. High-resolution X-ray photoelectron spectra (XPS) of (a) Co 2p, (b) V 2p and (c) N 1 s in Co₂V₂O₇@NF, N-Co₂V₂O₇@NF and N-Co₂VO₄/VO₂@NF, respectively.



Fig. 3. Electrocatalytic OER performance. (a) OER polarization curves, (b) Tafel plot, and (c) comparison of overpotential at 10 mA cm⁻² and Tafel of N-Co₂V₂O₂@NF with previously reported catalysts. (d) The turnover frequency (TOF) profiles, (e) C_{d1} plots, and (f) chronopotentiometry test in 1.0 M KOH (Inset: polarization curves before and after 7000 CV).

lysts, N-Co₂V₂O₇@NF also exhibits superior OER overpotential and Tafel slope (Fig. 3c and Table S3).

TOF values are calculated according to outcomes of inductively coupled plasma atomic emission spectrometry (ICP-AES) (Table S4). Fig. 3d demonstrates the TOF values of the five cobalt-vanadium catalysts at gradually increased applied voltage, and N-Co₂V₂O₇@NF possesses the highest TOF value at the same potential, reflecting its robust intrinsic activity [41,42]. Fig. S13 exhibits the TOF values of five cobalt vanadium-based catalysts at 300 mV overpotential. The intrinsic activity of the N-Co₂V₂O₇@NF catalyst (N content: 2.0%) improves about eight times compared with the precursor Co₂V₂O₇@NF and three times with N-Co₂V₂O₇@NF-300 (N content: 1.0%), suggesting that *N*-doping plays a crucial role in the intrinsic activity. The electrochemically active surface area (ECSA) and C_{dl} were also calculated (Fig. S14). Similarly, N-Co₂V₂O₇@NF displayed the largest ECSA and exposed



Fig. 4. Electrocatalytic HER performance. (a) HER polarization curves. (b) Tafel slopes. (c) Comparison of overpotential at 10 mA cm⁻² and Tafel value of N-Co₂VO₄/VO₂@NF with previously reported catalysts. (d) TOF profiles. (e) Nyquist plots (Impedances marked after catalysts are charge transfer resistances). (f) Chronopotentiometry curve (Inset: polarization curves initial and after 5000 CV).

the most active sites, about six times that of $Co_2V_2O_7@NF$ and N- $Co_2V_2O_7@NF$ -300 (Fig. 3e and S15). Interestingly, neither C_{d1} nor ECSA, $Co_2V_2O_7@NF$ -300 (30.9 mF cm⁻² and 772.5 cm²) has a slight preponderance over $Co_2V_2O_7@NF$ (28.6 mF cm⁻² and 715 cm²). To summarize, while the *N*-doping content is low, N will primarily reinforce the intrinsic activity, yet when more N is introduced, it strengthens the inherent activity and increases the number of active.

EIS was used to explore the charge transfer resistance (R_{ct}) relating to the conductivity of the catalyst. The five cobalt vanadium catalysts showed the following sequence: N-Co₂V₂O₇@NF (0.7 Ω) < N-Co₂VO₄/V₂O₃@NF (1.3 Ω) < N-Co₂VO₄/VO₂@NF (1.5 Ω) < Co₂V₂O₇@NF-300 (2.4 Ω) < Co₂V₂O₇@NF (3.1 Ω) (Fig. S16). Obviously, the electrode resistance is visibly reduced after introducing the N element, echoing previous literature [43,44]. In addition, the V can provide d electrons to Co to exhibit stronger catalytic activity [9]. Here, adding the N element can diminish the catalyst's resistance, thereby accelerating this process to ultimately strengthen the intrinsic activity and boost the catalyst's OER performance.

On account of the excellent performance of N-Co₂V₂O₇, a longterm stability test at a fixed current density of 100 mA cm⁻² was executed. The activity only degraded 1.3 % after 170 h (Fig. 3f). Moreover, the stability was also investigated using a multi-cycle CV test (inset: Fig. 3f), appearing that, after 7000 CV cycles, the polarization curve at 100 mA cm⁻² had decreased by 2.5 %. After the stability test, the nanosheet morphology remained intact (Fig. S17). At the same time, the chemical composition of the catalyst surface changed slightly, in which the proportions of Co³⁺ and V⁵⁺ increased (from 49.9 % to 67.3 % and 35.2 % to 46.6 %), while *N*-M decreased (from 31 % to 8.2 %). The high-valence states have resulted from the strong oxidation condition of OER (Fig. S18) [20,45].

The HER performance of the catalyst was investigated in a 1.0 M KOH solution. The polarization curves of catalysts have been shown in Fig. 4a. The N-Co₂VO₄/VO₂@NF can achieve an earlier onset potential and a faster current density increase with 87 mV overpotential at 10 mA cm⁻², being superior to Co₂V₂O₇@NF (208 mV), N-Co₂V₂O₇@NF-300 (206 mV), N-Co₂V₂O₇@NF (143 mV) and N-Co₂VO₄/V₂O₃@NF (137 mV) (Fig. S19). Notably,



Fig. 5. (a) Differential charge density diagram of N-Co₂V₂O₇ (The blue, yellow, red, gray and orange balls represent Co, V, N, O, and H atoms, respectively. Specially, the cyan region represents the depletion of electrons, and the yellow represents the accumulation of electrons.). (b) The density of states (DOS) of different catalysts. (c) OER mechanism illustration of N-Co₂V₂O₇. (d) Gibbs free energy diagrams during OER progress. (e) HER mechanism illustration of N-Co₂V₂O₄/VO₂. (d) Gibbs free energy diagrams during HER progress.

the N-Co₂VO₄/VO₂@NF outperforms Pt/C@NF when the current density exceeds 130 mA cm⁻² due to the unique nanosheet structure facilitating electrolyte diffusion and gas emission [46,47]. The Tafel slope corresponding to N-Co₂VO₄/VO₂@NF is 88 mV dec⁻¹, suggesting that its efficient HER kinetics follows the Volmer-Heyrovsky procedure: (1) M + e⁻ + H₂O \rightarrow M-H^{*} + OH⁻; (2) M-H^{*} + e⁻ + H₂O \rightarrow M + OH⁻ + H₂ (Fig. 4b) [48]. Besides, N-Co₂VO₄/VO₂@NF has superior HER activity to most previously reported HER electrocatalysts (Fig. 4c and Table S5).

TOF values also investigated the intrinsic activity of several HER catalysts. As shown in Fig. 4d, with the increase of the applied voltage, the TOF values of five cobalt vanadium-based electrocatalysts rise rapidly. N-Co₂VO₄/VO₂@NF is the most apparent among them, implying that each active site can release the most significant number of H₂ molecules per second [5]. The EIS shown in Fig. 4e demonstrates that the R_{ct} of N-Co₂VO₄/VO₂@NF is 3.8 Ω , second only to Pt/C@NF (1.5 Ω), implying high conductivity during the HER process.

Ultimately, long-term testing of N-Co₂VO₄/VO₂@NF was conducted and can stabilize for 700 h at a current density of 100 mA cm⁻² with minor deterioration (Fig. 4f). Furthermore, after 5000 CV cycle tests, the overpotential increased by only 2.7 % at 100 mA cm⁻² (inset: Fig. 4f). Notably, the morphology and chemical states of the catalyst did not change evidently after the stability test, conforming the superior stability of catalyst (Fig. S20-S21).

DFT calculations were performed to understand further the enhanced function of N doping toward electrocatalyst activity. After N doping, substantial charge redistribution occurs in N-Co₂V₂O₇ and N-Co₂V₂O₄/VO₂, and most of the charges accumulate around N, which favors the subsequent electron transfer and absorption/desorption intermediates (Fig. 5a and S22) [49]. To sight more about the origin of the high intrinsic activity after introducing the N element, DOS was employed. N-Co₂V₂O₇ and N-Co₂V₂O₄/VO₂ confirm significantly contracted band gaps, corresponding to reinforced electrical conductivity after *N*-doping (Fig. 5b) [50–52]. Meanwhile, N-Co₂V₂O₇ (–1.11 eV), suggesting

more favorable adsorption of active intermediate [53]. The OER mechanism and corresponding Gibbs free energy diagram are illustrated in Fig. 5c and 5d. First, OH⁻ loses an electron by attaching to the Co site; next, a proton is attacked by another OH⁻, and then deprotonates to form *OOH [54]. Finally, the proton-coupled electron transfer leads to the release of O₂ [55]. In these, the rate-determining step (RDS) is the second step (*OH + OH⁻ \rightarrow *O + H₂O + e⁻). The N-Co₂V₂O₇ shows the smallest reaction Gibbs free energy (1.29 eV) compared with 1.52 eV on Co₂V₂O₇, indicating that the electronic structure and d-band center of the catalyst can be controlled by doping N element into Co₂V₂O₇, resulting in a calculated free energy closer to the optimal value of 1.23 eV, thus devoting to improved OER performance (Table S6) [56].

Fig. 5e and 5f depict the HER mechanism and corresponding Gibbs free energy diagram, where N-Co₂V₂O₄/VO₂ display a clear shrinkage trend implying a more favorable HER pathway [57]. Typically, the activation of the H₂O toward the transition state (TS) serves as the RDS for HER, and N-Co₂V₂O₄/VO₂ exhibits a more desirable energy barrier of 1.44 eV than N-Co₂V₂O₇ (2.29 eV), inferring a synergistic effect at N-Co₂V₂O₄/VO₂ interface can significantly accelerate the absorption of H₂O (Table S7) [58]. Additionally, as diverse adsorption capacities of various sites for different intermediates, a hydrogen spillover phenomenon occurs on N-Co₂V₂O₄/VO₂ during the Heyrovsky step, which can regulate the activated site of H* to moderate hydrogen binding energy, 0.13 eV for N-Co₂V₂O₄/VO₂, which closes to the ideal value (0 eV) [59–61]. This noticeable difference can be attributed to the synergistic effects in N-Co₂V₂O₄/VO₂ interface and the hydrogen spillover phenomenon caused by N-doping, which can gear up H₂O-activation and balance the H^{*} adsorption and subsequent H₂ desorption, respectively, resulting in better HER performance.

Considering the extraordinary OER and HER activity of *N*-doped cobalt vanadium oxides, an OWS system was assembled by using N-Co₂V₂O₇@NF as anode and N-Co₂V₂O₄/VO₂@NF as the cathode (Fig. 6a). This device only needs a cell voltage of 1.52 V at 10 mA cm⁻² being equivalent to the RuO₂@NF⁽⁺⁾||Pt/C@NF⁽⁻⁾ (1.51 V) (Fig. 6b-c). Simultaneously, this N-Co₂V₂O₇@NF⁽⁺⁾||N-



Fig. 6. (a) Schematic diagram of OWS system. (b) Polarization curves of N-Co₂V₂O₇@NF⁽⁺⁾||N-Co₂VO₄/VO₂@NF⁽⁻⁾ and RuO₂@NF⁽⁻⁾||Pt/C@NF⁽⁻⁾ in 1.0 M KOH electrolyte. (c) Comparison of cell voltages for OWS system at 200, 300, and 1000 mA cm⁻². (d) Comparison of the alkaline OWS activity with previous reports at 10 mA cm⁻². (e) Chronopotentiometry tests at 100 and 500 mA cm⁻².

Co₂V₂O₄/VO₂@NF⁽⁻⁾ OWS system was compared to the newly reported cell voltage of the two-electrode devices at 10 mA cm⁻², suggesting that it outperforms most systems (Fig. 6d). In addition, once the applied voltage exceeds 1.75 V, N-Co₂V₂O₇@NF⁽⁺⁾||N- Co₂-V₂O₄/VO₂@NF⁽⁻⁾ requires less voltage to achieve the same current density than RuO₂@NF⁽⁺⁾||Pt/C@NF⁽⁻⁾. Moreover, this water electrolysis device can be stabilized for 120 h at 100 mA cm⁻² with a voltage loss of 0.11 % and only changed 1.8 % after 60 h at 500 mA cm⁻² (Fig. 6e-f).

Based on the above discussions, this excellent OER activity of the N-Co₂V₂O₇@NF catalyst can be ascribed to the following factors: (1) Vertical nanosheet arrangement can speed up electrolyte transport and gas emissions. (2) The combination of cobalt and vanadium bimetal can enhance the activity of metal sites. (3) *N*-doping can promote charge transfer and regulate the adsorption of intermediates. For HER, *N* doping is only part of the reason for improving performance, and the formation of the N-Co₂V₂O₄/VO₂ interface is also a non-negligible reason.

4. Conclusion

In summary, we fabricated cobalt vanadium oxide nanosheet series by hydrothermal and nitrification processes. The results revealed that the nitrification temperature significantly influenced the catalysts' microstructure and crystallinity. The maximum Ndoping (about 2 %) was achieved after nitrification temperature exceeded 350 °C. Electrochemical tests found that N-Co₂V₂O₇@NF prepared at 350 °C and N-Co₂VO₄/VO₂@NF prepared at 400 °C had the best OER and HER electrocatalytic activities and stabilities among all catalysts, respectively. Impressively, overall water splitting assembled by N-Co₂V₂O₇@NF⁽⁺⁾||N-Co₂VO₄/VO₂@NF⁽⁻⁾ can achieve a cell voltage of 1.93 V@500 mA cm⁻² even superior to $RuO_2@NF^{(+)}||Pt/C@NF^{(-)}$ (2.02 V). The DFT study indicates that Ndoping can reduce the band gap of materials and promote the adsorption/desorption of intermediates, resulting in faster charge transfer and smaller reaction energy barriers. Notably, hydrogen spillover occurs at the N-Co₂VO₄/VO₂ interface, leading to a more desirable binding energy for H_{ad}, making the HER performance more ideal. This work provides a feasible strategy for constructing efficient *N*-doped electrocatalysts for highly industrial applications.

CRediT authorship contribution statement

Zuyang Luo: Conceptualization, Writing – original draft. Qimin Peng: Data curation. Zhiyang Huang: Visualization. Lixia Wang: Investigation. Yuting Yang: Software. Jiaxin Dong: Data curation. Tayirjan Taylor Isimjan: Writing – review & editing. Xiulin Yang: Supervision, Writing – review & editing.

Data availability

Data will be made available on request.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Data availability

The data that support the findings of this study are available from the corresponding author upon request.

Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.jcis.2022.09.069.

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